

# Infrared spectroscopic detection of the disilenyl ( $\text{Si}_2\text{H}_3$ ) and d3-disilenyl ( $\text{Si}_2\text{D}_3$ ) radicals in silane and d4-silane matrices

David Sillars<sup>a</sup>, Chris J. Bennett<sup>a</sup>, Yoshihiro Osamura<sup>b</sup>, Ralf I. Kaiser<sup>a,\*</sup>

<sup>a</sup> Department of Chemistry, University of Hawai'i at Manoa, 2545 The Mall, Honolulu, HI 96822, USA

<sup>b</sup> Department of Chemistry, Rikkyo University, 3-34-1 Nishi-ikebukuro, Tokyo 171-8501, Japan

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## Abstract

The disilenyl,  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$ , and the d3-isotopomer were detected for the first time via infrared spectroscopy in low temperature silane matrices upon an irradiation of the sample matrices with energetic electrons. The  $\nu_5$  fundamental was observed at 651 and 493  $\text{cm}^{-1}$ , respectively. In the d4-silane matrix, the  $\nu_4$  at 683  $\text{cm}^{-1}$  was noticed, too. Our investigations suggest that this radical is formed via radiolysis of silylsilylene,  $\text{H}_3\text{SiSiH}(\text{X}^1\text{A}')$ , and disilene,  $\text{H}_2\text{SiSiH}_2(\text{X}^1\text{A}_g)$ . The new absorption of the  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  radical may be employed in future spectroscopic monitoring of chemical vapor deposition processes and in astronomical searches of silicon-bearing molecules toward the carbon star IRC + 10216.

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## 1. Introduction

In recent years, the structures, energetics, and spectroscopic properties of small hydrogenated silicon clusters of the generic formula  $\text{Si}_x\text{H}_y$  ( $x = 1, 2$ ;  $y = 1-6$ ) attracted considerable attention [1–6]. Due to their technological applications such as silane chemical vapor deposition and semiconductor processing, a detailed understanding of the structures and energetics of these clusters is also of fundamental interest in solid state chemistry and physics [7]. Here, plasma etching processes, reactive plasmas, and chemical vapor deposition techniques are of wide technological interest to produce silicon-bearing nano-particles and also amorphous silicon films (a-Si:H) [8]. The latter have fundamental applications in material sciences, particularly in the development of solar cells, electro-photographic drums, and arrays for liquid crystal displays. In chemical vapor deposition processes, silane ( $\text{SiH}_4$ ) or disilane ( $\text{Si}_2\text{H}_6$ ) are decomposed by hot filaments or electrons generated within plasmas [9]. During these procedures, silicon-bearing species such as  $\text{SiH}_y$  ( $y = 1-3$ ) and clusters like

$\text{Si}_2\text{H}_y$  ( $y = 1-5$ ) have been identified in the gas phase as major growth species to produce amorphous, often porous silicon films [10–12].

Identifying the controlling and growth-limiting reactions for the production of amorphous silicon films requires a rigorous knowledge of the time-dependent concentrations of silicon-bearing species in actual chemical vapor deposition processes. The in situ characterization of gaseous species in CVD processes is predominantly carried out via time resolved threshold ionization mass spectrometry such as of the disilyl radical,  $\text{Si}_2\text{H}_5(\text{X}^2\text{A}')$  [9,13–15]. Despite the importance of the  $\text{Si}_2\text{H}_y$  ( $y = 1-5$ ) species in chemical vapor deposition processes, no time resolved spectroscopic probes have been established. Here, a detailed knowledge of the infrared absorption features might help to follow the chemical evolution of silicon CVD process not only via mass spectrometry, but also through time resolved infrared spectroscopy. So far, the most intense vibrational levels of disilane,  $\text{Si}_2\text{H}_6(\text{X}^1\text{A}_{1g})$  [2158  $\text{cm}^{-1}$ , 841–835  $\text{cm}^{-1}$ ] [16], of the disilyl radical,  $\text{Si}_2\text{H}_5(\text{X}^2\text{A}')$  [843  $\text{cm}^{-1}$ ] [17], of two  $\text{Si}_2\text{H}_4$  isomers disilene ( $\text{H}_2\text{SiSiH}_2$ ,  $\text{X}^1\text{A}_g$ ) [904, 2180, 2201  $\text{cm}^{-1}$ ] [18] and silylsilylene ( $\text{HSiSiH}_3$ ,  $\text{X}^1\text{A}'$ ) [707, 861–857, 1963, 2112, 2154–2145, 2167  $\text{cm}^{-1}$ ] [16,18], and of the dibridged  $\text{Si}_2\text{H}_2$  species [1093–1100

\* Corresponding author. Fax: +8089565908.

E-mail address: [kaiser@gold.chem.hawaii.edu](mailto:kaiser@gold.chem.hawaii.edu) (R.I. Kaiser).

$\text{cm}^{-1}$ ] [16] have been determined experimentally in low temperature matrices. It is fairly surprising that no infrared spectroscopic, but only a Fourier transform microwave investigation on any  $\text{Si}_2\text{H}_3$  isomers has been carried out to date [19].

Our investigations are also valuable to contribution to a detection of hitherto unidentified silicon-bearing molecules in the circumstellar envelopes of the carbon star IRC + 10216. So far, seven silicon containing species, i.e., silicon carbide (SiC), silicon nitride (SiN), silicon cyanide (SiCN), and tetracarbonylsilicate (CCCCSi) together with two cyclic molecules ( $\text{SiC}_2$ ,  $\text{SiC}_3$ ) have been detected in this stellar outflow [20,21]. Here, the silane molecule – a potential precursor to complex organo-silicon molecules – has also been detected in the outflow of IRC + 10216 via infrared telescopes [22].

In this Letter, we present a combined experimental and theoretical study on the infrared spectroscopic detection of the  $\text{Si}_2\text{H}_3$  radical and its deuterated counterpart in low temperature silane and d4-silane matrices. We have demonstrated earlier that in low temperature methane matrices, energetic electrons induce primarily a carbon–hydrogen bond cleavage to form a methyl radical and atomic hydrogen ( $\text{CH}_3(\text{X}^2\text{A}_2'') + \text{H}(^2\text{S}_{1/2})$ ). Neighboring methyl radicals were found to recombine to an ethane molecule,  $\text{C}_2\text{H}_6(\text{X}^1\text{A}_{1g})$ . A consecutive irradiation of the sample with higher electron currents lead to a successive loss of hydrogen atoms via the reaction sequence  $\text{C}_2\text{H}_6(\text{X}^1\text{A}_{1g}) \rightarrow \text{C}_2\text{H}_5(\text{X}^2\text{A}') \rightarrow \text{C}_2\text{H}_4(\text{X}^1\text{A}_g) \rightarrow \text{C}_2\text{H}_3(\text{X}^2\text{A}') \rightarrow \text{C}_2\text{H}_2(\text{X}^1\Sigma_g^+)$ . Recently, a similar approach was utilized to form the disilyl radical,  $\text{Si}_2\text{H}_5(\text{X}^2\text{A}')$ , in silane matrices. In this Letter, we expand this protocol and synthesize the  $\text{Si}_2\text{H}_3$  radical via  $\text{Si}_2\text{H}_4$  isomers in  $\text{SiH}_4$  and  $\text{SiD}_4$  ices,  $\text{Si}_2\text{H}_6(\text{X}^1\text{A}_{1g}) \rightarrow \text{Si}_2\text{H}_5(\text{X}^2\text{A}') \rightarrow \text{Si}_2\text{H}_4$  isomers  $\rightarrow \text{Si}_2\text{H}_3$  isomers. These studies provide the position of the most intense, hitherto elusive infrared absorption frequencies of these species. Complete descriptions of the  $\text{Si}_2\text{H}_3$  and  $\text{Si}_2\text{H}_4$  potential energy surfaces are given in a forthcoming publication.

## 2. Experimental

The laboratory experiments were conducted in an ultrahigh vacuum (UHV) chamber. Briefly, the vessel consists of a cylindrical stainless steel chamber which can be pumped down to  $2 \times 10^{-10}$  Torr by a magnetically suspended turbopump backed by an oil-free scroll pump. A two stage closed cycle helium refrigerator is interfaced to the machine carrying a polished silver crystal. Cooled to  $10.4 \pm 0.2$  K, the latter serves as a substrate for the ice condensate. The silane and d4-silane ice were prepared at 10 K by condensing silane (99.99%) and d4-silane (99.99%) at pressures of  $5\text{--}6 \times 10^{-8}$  Torr for 30 min onto the silver target. A

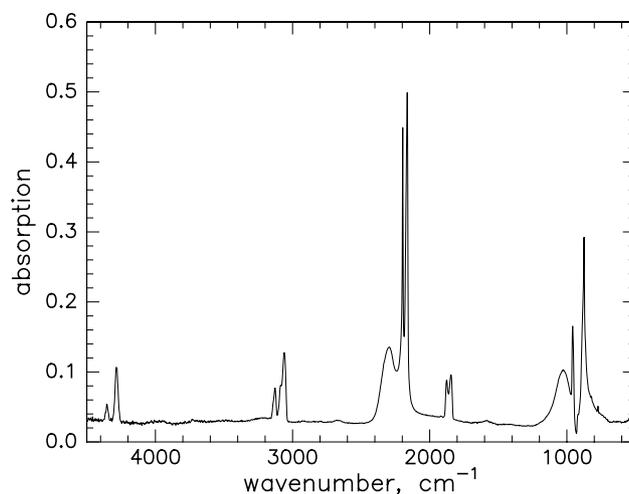


Fig. 1. Infrared spectrum of the silane frost at 10 K. The assignments of the peaks are compiled in Table 1.

comparison of our data (Fig. 1 and Table 1) with the literature data suggests that the frozen high temperature modifications of  $\text{SiH}_4(\text{I})$  and  $\text{SiD}_4(\text{I})$  dominate the constitution of the sample. However, the presence of a  $\nu_2$  peak at  $957 \text{ cm}^{-1}$  which is infrared inactive in crystalline  $\text{SiH}_4(\text{I})$  suggests that the sample is partially disordered, possibly amorphous. We would like to stress that  $\text{SiH}_4(\text{II})$  can be excluded because we do not observe any fine structure in  $\nu_3$  ( $2163 \text{ cm}^{-1}$ ) and  $\nu_4$  ( $876 \text{ cm}^{-1}$ ). To determine the ice thickness, the infrared absorption features at  $2163$  and  $876 \text{ cm}^{-1}$  were integrated; the ice thickness was calculated then via the Lambert–Beer relationship [23]. Considering the integrated absorption coefficients of these fundamentals, i.e.,  $2.5 \times 10^{-17}$  and  $2.0 \times 10^{-17} \text{ cm}$ , and a density of the silane ice of

Table 1

Infrared absorptions of the silane frosts (sh: shoulder);  $\alpha$ ,  $\beta$ , and  $\gamma$  denote lattice modes of the silane sample

| Frequency ( $\text{cm}^{-1}$ ) | Assignment                     |
|--------------------------------|--------------------------------|
| 4354                           | $2 \nu_3$                      |
| 4283                           | $\nu_1 + \nu_3$                |
| 3142 (sh)                      | $\nu_3 + \nu_4 + \beta$        |
| 3128                           | $\nu_2 + \nu_3$                |
| 3087                           | $\nu_3 + \nu_4 + \alpha$       |
| 3060                           | $\nu_3 + \nu_4$                |
| 2300                           | $\nu_3 + \gamma$               |
| 2250                           | $\nu_3 + \beta$                |
| 2195                           | $\nu_3 + \alpha$               |
| 2163                           | $\nu_3$                        |
| 1876                           | $\nu_2 + \nu_4 + \alpha$       |
| 1844                           | $\nu_2 + \nu_4$                |
| 1050                           | $\nu_2 + \beta/\nu_4 + \gamma$ |
| 957                            | $\nu_2$                        |
| 948                            | $\nu_4 + \beta$                |
| 918                            | $\nu_4 + \alpha$               |
| 876                            | $\nu_4$                        |

$0.77 \pm 0.03 \text{ g cm}^{-3}$  [24], an optical thickness of  $0.2 \text{ }\mu\text{m}$  silane was derived.

These ices were irradiated at 10 K with 5 keV electrons generated in an electron gun at beam currents of 1000 nA by wobbling the electron beam over the target area of  $3.0 \pm 0.4 \text{ cm}^2$ . Accounting for irradiation times of between of 120 min and the extraction efficiency of 78.8% of the electrons, this exposes the targets to  $4.5 \times 10^{16}$  electrons. To guarantee an identification of the reaction products in the ices a Fourier transform infrared spectrometer was utilized. The Nicolet 510 DX FTIR machine ( $5000\text{--}500 \text{ cm}^{-1}$ ) operated in an absorption–reflection–absorption mode (reflection angle  $\alpha = 75^\circ$ ) (resolution  $0.5\text{--}2 \text{ cm}^{-1}$ ). The infrared beam was coupled via a mirror flipper outside the spectrometer, passed through a differentially pumped potassium bromide (KBr) window, was attenuated in the ice sample prior and after reflection at a polished silver waver, and exited the main chamber through a second differentially pumped KBr window before being monitored via a liquid nitrogen cooled detector.

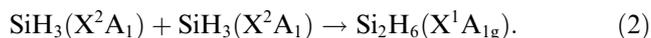
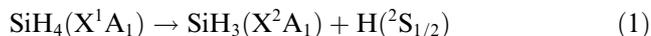
### 3. Theoretical approach

The molecular structures of various isomers for the  $\text{Si}_2\text{H}_2$ ,  $\text{Si}_2\text{H}_3$ , and  $\text{Si}_2\text{H}_4$  species were optimized in terms of ab initio density functional B3LYP methods [25,26] with the 6-311G(d,p) basis set [27]. Most of the structures of  $\text{Si}_2\text{H}_3$  and  $\text{Si}_2\text{H}_4$  have been studied previously [28,19], but we have calculated the vibrational frequencies and infrared intensities for these  $\text{Si}_2\text{H}_n$  species and their deuterated  $\text{Si}_2\text{D}_n$  counterparts to identify the newly formed silicon-bearing species unambiguously. The coupled cluster CCSD(T) calculations [29,30] with the aug-cc-pVTZ basis set [31] were also performed at the optimized structures obtained with the B3LYP method in order to compare the relative energies of various isomers. All computations were carried out using the GAUSSIAN 98 program package [32]. The relative energies stated in the text are the values obtained with the CCSD(T) method corrected with the zero-point vibrational energies obtained with the B3LYP method (Tables 2 and 3).

### 4. Results

Our spectroscopic studies suggest that the response of the silane system upon the electron bombardment is governed by an initial formation of the silyl radical,  $\text{SiH}_3(\text{X}^2\text{A}_1)$ , plus atomic hydrogen. Here, absorption features of the silyl and d3-silyl radicals are observable immediately at  $722 \text{ cm}^{-1}$  ( $\nu_2$ ;  $\text{SiH}_3$ ) as well as  $540 \text{ cm}^{-1}$  ( $\nu_2$ ;  $\text{SiD}_3$ ) and  $669 \text{ cm}^{-1}$  ( $\nu_4$ ;  $\text{SiD}_3$ ) [17]. The  $\nu_2$  peak agrees well with previous matrix studies [33–35]. Strong

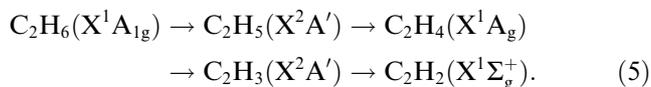
absorptions at  $820 \text{ cm}^{-1}$  ( $\nu_6$ ;  $\text{Si}_2\text{H}_6(\text{X}^1\text{A}_{1g})$ ) and  $935 \text{ cm}^{-1}$  ( $\nu_8$ ;  $\text{Si}_2\text{H}_6(\text{X}^1\text{A}_{1g})$ ) as well as at  $606 \text{ cm}^{-1}$  ( $\nu_6$ ;  $\text{Si}_2\text{D}_6(\text{X}^1\text{A}_{1g})$ ) and  $1541 \text{ cm}^{-1}$  ( $\nu_5$ ;  $\text{Si}_2\text{D}_6(\text{X}^1\text{A}_{1g})$ ) were observable, too. Mechanical studies suggested that the disilane molecule was formed via recombination of two neighboring silyl radicals (reactions (1) and (2)) [17]:



We would like to stress that the assignment of all peaks was double checked in perdeuterated silane ( $\text{SiD}_4$ ) matrices. Here, we detected two fundamentals of the d3-silyl radical,  $\text{SiD}_3(\text{X}^2\text{A}_1)$ , at  $669 \text{ cm}^{-1}$  ( $\nu_4$ ; shoulder) and  $540 \text{ cm}^{-1}$  ( $\nu_2$ ; Fig. 3); this assignment corresponds very well with previously assigned peak positions at  $668 \text{ cm}^{-1}$  and about  $546 \text{ cm}^{-1}$  in noble gas matrices. Likewise, the d6-disilane molecule was observed in our experiments through its absorptions at  $606 \text{ cm}^{-1}$  ( $\nu_6$ ; Fig. 2) and  $1541 \text{ cm}^{-1}$  ( $\nu_5$ ). Summarized, at low irradiation currents of 10 nA, solely absorptions of the silyl radical,  $\text{SiH}_3(\text{X}^2\text{A}_1)$ , and of the disilane molecule,  $\text{Si}_2\text{H}_6(\text{X}^1\text{A}_{1g})$ , together with their deuterated counterparts can be observed unambiguously in the silane matrix (Table 4). As the irradiation time increases, previous studies shows also the formation of the disilyl radical,  $\text{Si}_2\text{H}_5(\text{X}^2\text{A}')$ , via reaction (3). Here, absorptions could be detected at  $843 \text{ cm}^{-1}$  ( $\nu_6$ ;  $\text{Si}_2\text{H}_5(\text{X}^2\text{A}')$ ) and  $621 \text{ cm}^{-1}$  ( $\nu_6$ ;  $\text{Si}_2\text{D}_5(\text{X}^2\text{A}')$ )



Having identified the disilyl radical as a radiolysis product of the disilane molecule, we investigate now how the system responds on significantly enhanced exposure to electrons. As the irradiation time increases, new absorptions appear at  $867 \text{ cm}^{-1}$  (silane matrix; Fig. 2) and  $635 \text{ cm}^{-1}$  (d4-silane matrix, Fig. 2). A shoulder is also observable at  $528 \text{ cm}^{-1}$  in the d4-silane matrix. Since our goal is to observe the reaction sequence (4) in a similar manner as carried out previously in methane samples (reaction sequence (5)), we compare the observed absorptions with those computed for the three energetically most stable  $\text{Si}_2\text{H}_4$  isomers (Table 2):



Indeed, the  $867$  and  $635 \text{ cm}^{-1}$  modes can be clearly identified as the most intense  $\nu_5$  fundamental (umbrella mode of the  $\text{SiH}_3/\text{SiD}_3$  group) of the  $\text{H}_3\text{SiSiH}(\text{X}^1\text{A}')$  isomer and its isotopomer. The  $528 \text{ cm}^{-1}$  peak correlates nicely with the Si–D-bending mode of the  $\text{D}_3\text{SiSiD}(\text{X}^1\text{A}')$  species (Table 2). Scaling the computed frequencies with a factor of 0.99 yields an excellent

Table 2

Unscaled vibrational frequencies ( $\text{cm}^{-1}$ ) and their infrared intensities ( $\text{km mol}^{-1}$ ) of the three lowest energy  $\text{Si}_2\text{H}_4$  and  $\text{Si}_2\text{D}_4$  isomers calculated with B3LYP/6-311G(d,p) method

| Symmetry         | Frequency                                    | Intensity | Frequency                  | Intensity | Characterization          |
|------------------|--|-----------|----------------------------|-----------|---------------------------|
| $C_{2h}$         | <b>1a</b> $\text{H}_2\text{SiSiH}_2(X^1A_g)$ |           | $\text{D}_2\text{SiSiD}_2$ |           |                           |
| $\nu_1 (a_g)$    | 2229   | 0         | 1596                       | 0         | Sym. SiH str.             |
| $\nu_2 (a_g)$    | 956  | 0         | 713                        | 0         | Sym. $\text{SiH}_2$ bend  |
| $\nu_3 (a_g)$    | 562  | 0         | 614                        | 0         | Si–Si str.                |
| $\nu_4 (a_g)$    | 324  | 0         | 254                        | 0         | Rock                      |
| $\nu_5 (a_u)$    | 2258   | 148       | 1634                       | 79        | Asym. SiH str.            |
| $\nu_6 (a_u)$    | 525  | 0         | 371                        | 0         | Torsion                   |
| $\nu_7 (a_u)$    | 348  | 23        | 249                        | 12        | Deform                    |
| $\nu_8 (b_g)$    | 2246   | 0         | 1625                       | 0         | Asym. SiH str.            |
| $\nu_9 (b_g)$    | 616  | 0         | 469                        | 0         | Deform                    |
| $\nu_{10} (b_u)$ | 2225   | 110       | 1590                       | 61        | Sym. SiH str.             |
| $\nu_{11} (b_u)$ | 920  | 183       | 665                        | 95        | Sym. $\text{SiH}_2$ bend  |
| $\nu_{12} (b_u)$ | 447  | 32        | 327                        | 17        | Rock                      |
| $C_s$            | <b>1b</b> $\text{H}_3\text{SiSiH}(X^1A')$    |           | $\text{D}_3\text{SiSiD}$   |           |                           |
| $\nu_1 (a')$     | 2214   | 128       | 1597                       | 75        | SiH str.                  |
| $\nu_2 (a')$     | 2179   | 76        | 1553                       | 40        | Sym. SiH str.             |
| $\nu_3 (a')$     | 2037   | 206       | 1465                       | 106       | SiH str.                  |
| $\nu_4 (a')$     | 933  | 71        | 671                        | 60        | $\text{SiH}_3$ deform     |
| $\nu_5 (a')$     | 868  | 222       | 641                        | 78        | $\text{SiH}_3$ umbrella   |
| $\nu_6 (a')$     | 716  | 62        | 533                        | 42        | SiH bend                  |
| $\nu_7 (a')$     | 426  | 24        | 377                        | 16        | Rock                      |
| $\nu_8 (a')$     | 368  | 8         | 288                        | 9         | Si–Si str.                |
| $\nu_9 (a'')$    | 2188   | 109       | 1580                       | 63        | Asym. SiH str.            |
| $\nu_{10} (a'')$ | 957  | 38        | 686                        | 21        | $\text{SiH}_3$ deform     |
| $\nu_{11} (a'')$ | 386  | 29        | 280                        | 15        | Deform                    |
| $\nu_{12} (a'')$ | 51   | 10        | 37                         | 5         | Torsion                   |
| $C_1$            | <b>1c</b> $\text{H}_2\text{SiHSiH}(X^1A)$    |           | $\text{D}_2\text{SiDSiD}$  |           |                           |
| $\nu_1 (a)$      | 2220   | 127       | 1604                       | 72        | Asym. $\text{SiH}_2$ str. |
| $\nu_2 (a)$      | 2196   | 148       | 1573                       | 82        | Sym. $\text{SiH}_2$ str.  |
| $\nu_3 (a)$      | 2041   | 178       | 1468                       | 90        | SiH str.                  |
| $\nu_4 (a)$      | 1627   | 117       | 1168                       | 59        | Bridge H str.             |
| $\nu_5 (a)$      | 1009   | 356       | 732                        | 185       | Bridge H shift            |
| $\nu_6 (a)$      | 962  | 38        | 688                        | 7         | $\text{SiH}_2$ scissor    |
| $\nu_7 (a)$      | 858  | 88        | 615                        | 43        | Bridge H deform           |
| $\nu_8 (a)$      | 703  | 37        | 538                        | 38        | Rock                      |
| $\nu_9 (a)$      | 636  | 7         | 453                        | 2         | Twist                     |
| $\nu_{10} (a)$   | 477  | 27        | 442                        | 11        | Si–Si str.                |
| $\nu_{11} (a)$   | 448  | 3         | 344                        | 7         | Deform                    |
| $\nu_{12} (a)$   | 396  | 6         | 285                        | 4         | Rock                      |

agreement with our experimental observations. The computed integral absorption coefficients also suggest that the  $\nu_5$  band of  $\text{H}_3\text{SiSiH}(X^1A')$  is the strongest one. The remaining absorptions, e.g., the Si–H and Si–D stretching modes, are either obscured by the silane matrices or are too weak to be detectable. It is important to note that at 10 K, we were not able to detect any absorption of the thermodynamically more stable  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  isomer. Our calculations suggest that the  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  structure is favorable by about 25  $\text{kJ mol}^{-1}$  (27  $\text{kJ mol}^{-1}$  is the value with CCSD(T) method) compared to  $\text{H}_3\text{SiSiH}(X^1A')$ ; this agrees well with a recent study which obtained a value of 27.3

$\text{kJ mol}^{-1}$  [19]. Here, the most intense  $\nu_{11}$  mode of  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  lies within the  $\nu_4$  and  $\nu_4 + \alpha$  absorptions of the silane matrix (Table 1). Also, we would like to stress that the thermodynamically even less stable, hydrogen bridged  $\text{H}_2\text{SiHSiH}(X^1A)$  isomer is not detectable either (we computed that the  $\text{H}_3\text{SiSiH}(X^1A')$  structure is favored by about 3  $\text{kJ mol}^{-1}$  compared to the  $\text{H}_2\text{SiHSiH}(X^1A)$  isomer; again, this order of magnitude compares well with Sari's et al. data of 3  $\text{kJ mol}^{-1}$ ). If it existed, the  $\nu_3$  mode should be observable in our experiment. Therefore, we can conclude that  $\text{H}_2\text{SiHSiH}(X^1A)$  is not formed. This correlates nicely with the failed detection of the hydrogen-bridged

Table 3

Unscaled vibrational frequencies ( $\text{cm}^{-1}$ ) and their infrared intensities ( $\text{km mol}^{-1}$ ) of the three lowest energy  $\text{Si}_2\text{H}_3$  and  $\text{Si}_2\text{D}_3$  isomers calculated with B3LYP/6-311G(d,p) method

| Symmetry      | Frequency  | Intensity | Frequency                | Intensity | Characterization       |
|---------------|--|-----------|--------------------------|-----------|------------------------|
| $C_1$         | <b>2a</b> $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$   |           | $\text{D}_2\text{SiSiD}$ |           |                        |
| $\nu_1$ (a)   | 2210   | 121       | 1596                     | 68        | Asym. Si–H str.        |
| $\nu_2$ (a)   | 2180   | 109       | 1562                     | 57        | Sym. Si–H str.         |
| $\nu_3$ (a)   | 2041   | 182       | 1468                     | 92        | Si–H str.              |
| $\nu_4$ (a)   | 965  | 93        | 702                      | 42        | $\text{SiH}_2$ scissor |
| $\nu_5$ (a)   | 683  | 24        | 511                      | 18        | Rock                   |
| $\nu_6$ (a)   | 448  | 6         | 422                      | 7         | Deform                 |
| $\nu_7$ (a)   | 402  | 9         | 298                      | 4         | Out-of-plane           |
| $\nu_8$ (a)   | 385  | 13        | 279                      | 7         | Deform                 |
| $\nu_9$ (a)   | 192  | 7         | 142                      | 4         | Torsion                |
| $C_s$         | <b>2b</b> $\text{H}_3\text{SiSi}(\text{X}^2\text{A}'')$  |           | $\text{D}_3\text{SiSi}$  |           |                        |
| $\nu_1$ (a')  | 2188   | 115       | 1573                     | 68        | Asym. Si–H str.        |
| $\nu_2$ (a')  | 2155   | 74        | 1540                     | 39        | Sym. Si–H str.         |
| $\nu_3$ (a')  | 953  | 68        | 684                      | 44        | Si–H str.              |
| $\nu_4$ (a')  | 866  | 254       | 641                      | 112       | $\text{SiH}_2$ scissor |
| $\nu_5$ (a')  | 407  | 18        | 382                      | 19        | $\text{SiH}_3$ deform  |
| $\nu_6$ (a')  | 295  | 27        | 226                      | 17        | Sym. bend              |
| $\nu_7$ (a'') | 2191   | 115       | 1583                     | 65        | Asym. bend             |
| $\nu_8$ (a'') | 911  | 44        | 654                      | 23        | Deform                 |
| $\nu_9$ (a'') | 386  | 12        | 286                      | 6         | Rock                   |
| $C_s$         | <b>2c</b> $\text{H}_2\text{SiHSi}(\text{X}^2\text{A}'')$ |           | $\text{D}_2\text{SiDSi}$ |           |                        |
| $\nu_1$ (a')  | 2179   | 157       | 1559                     | 88        | Sym. Si–H str.         |
| $\nu_2$ (a')  | 1662   | 109       | 1193                     | 54        | Bridge H str.          |
| $\nu_3$ (a')  | 972  | 365       | 711                      | 176       | Scissor                |
| $\nu_4$ (a')  | 930  | 57        | 662                      | 23        | H bend                 |
| $\nu_5$ (a')  | 470  | 24        | 459                      | 30        | Si–Si str.             |
| $\nu_6$ (a')  | 425  | 22        | 319                      | 13        | $\text{SiH}_2$ rock    |
| $\nu_7$ (a'') | 2189   | 125       | 1583                     | 71        | Asym. Si–H str.        |
| $\nu_8$ (a'') | 726  | 24        | 519                      | 13        | Twist                  |
| $\nu_9$ (a'') | 407  | 7         | 302                      | 4         | Deform                 |

$\text{HSiHSiH}_3$  isomer which cannot be formed by a Si–H bond rupture from disilane.

As the irradiation time increases even further, we detected additional bands at  $651 \text{ cm}^{-1}$  (Fig. 3; silane target) as well as  $493$  and  $683 \text{ cm}^{-1}$  (Fig. 3; d4-silane). These absorptions cannot be attributed to any  $\text{Si}_2\text{H}_x/\text{Si}_2\text{D}_x$  ( $x = 4-6$ ) isomer. Instead, we compared these data with the energetically most stable  $\text{Si}_2\text{H}_3/\text{Si}_2\text{D}_3$  structures – assuming that the newly observed bands originate from an electron-induced decomposition of the  $\text{Si}_2\text{H}_4/\text{Si}_2\text{D}_4$  isomers. Here, we can verify the existence

of the  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  isomer together with its perdeuterated isotopomers via the  $\nu_5$  (rocking) and  $\nu_4$  ( $\text{SiH}_2$  scissor; shoulder) fundamentals. A scaling factor of about 0.97 converts the computed frequencies to the experimentally observed ones. In case of  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$ , the more intense  $\nu_1-\nu_4$  modes overlap with those of the silane molecule; likewise, the  $\nu_1-\nu_3$  modes of  $\text{D}_2\text{SiSiD}(\text{X}^2\text{A})$  cannot be observed (Table 4). Therefore, not only the microwave transitions [19], but also the infrared absorption can be utilized in future spectroscopic searches of the  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  radical. Compared to the

Table 4

Compilation of newly observed  $\text{Si}_2\text{H}_4$ ,  $\text{Si}_2\text{H}_3$ , and  $\text{Si}_2\text{H}_2$  species together with their deuterated counterparts in low temperature silane matrices

| Species                                      | Frequency ( $\text{cm}^{-1}$ ) | Fundamental | Frequency ( $\text{cm}^{-1}$ ) | Species                                      |
|--|--------------------------------|-------------|--------------------------------|--|
| $\text{H}_3\text{SiSiH}$                     | 867                            | $\nu_5$     | 635                            | $\text{D}_3\text{SiSiD}$                     |
| $\text{H}_3\text{SiSiH}$                     | –                              | $\nu_6$     | 528 (sh)                       | $\text{D}_3\text{SiSiD}$                     |
| <i><math>\text{H}_2\text{SiSiH}_2</math></i> | 898                            | $\nu_{11}$  | 658                            | <i><math>\text{D}_2\text{SiSiD}_2</math></i> |
| $\text{H}_2\text{SiSiH}$                     | 651; 641(sh)                   | $\nu_5$     | 493                            | $\text{D}_2\text{SiSiD}$                     |
|  |                                | $\nu_4$     | 683(sh)                        |  |

Absorptions in italics were observed after the sublimation of the silane matrices (sh: shoulder).

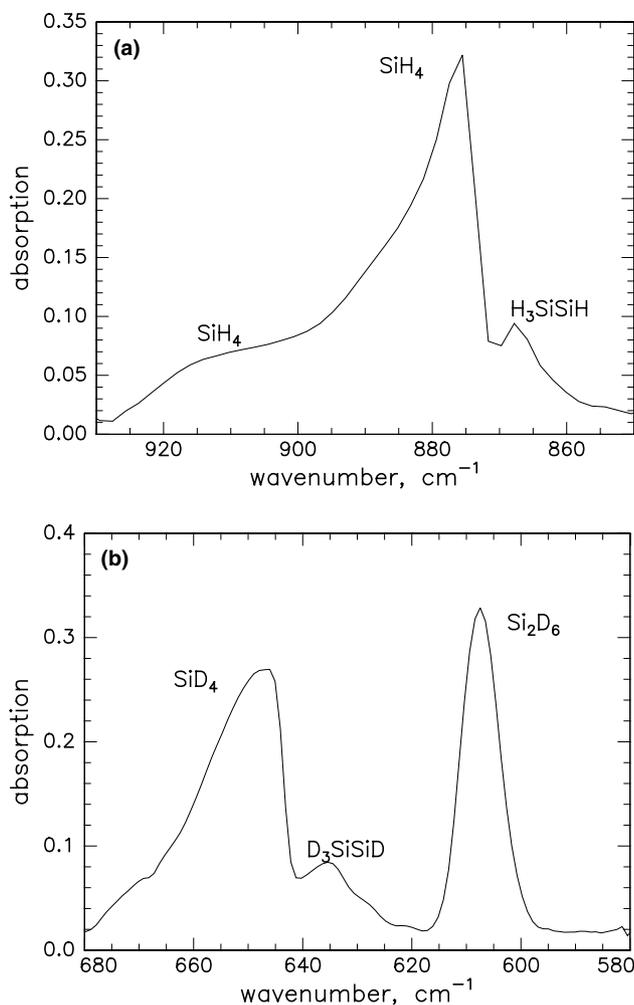


Fig. 2. New absorption features of the silylsilylene ( $\text{HSiSiH}_3$ ,  $X^1A'$ ) isomer at  $867\text{ cm}^{-1}$  (a) and of the d4-silylsilylene ( $\text{DSiSiD}_3$ ,  $X^1A'$ ) isomer at  $635\text{ cm}^{-1}$  (b) in the silane and d4-silane matrices, respectively, at 10 K (a); the  $\nu_4$  fundamentals at  $876$  and  $647\text{ cm}^{-1}$  as well as the  $\nu_4 + \alpha$  combination bands ( $918\text{ cm}^{-1}$ ) are also shown.

estimation by Sari's et al., which suggested that the bridged  $\text{H}_2\text{SiHSi}(X^2A'')$  isomer is the energetically most stable ones followed by  $\text{H}_2\text{SiSiH}(X^2A)$  ( $+1\text{ kJ mol}^{-1}$ ) and  $\text{H}_3\text{SiSi}(X^2A'')$  ( $+13\text{ kJ mol}^{-1}$ ), our investigation suggests that all three isomers range within  $0.5\text{ kJ mol}^{-1}$ . Note that we have used aug-cc-pVTZ basis set which involves diffuse functions and is larger than the ones by Sari's et al. Although they have estimated the relative energies by using coupled cluster method including full triple excitations, our values with CCSD(T) method would be reasonably accurate within the accuracy of  $\pm 5\text{ kJ mol}^{-1}$ . The molecular structures are almost identical with Sari's paper. The vibrational frequencies obtained by Sari et al. are overestimated due to the smaller basis set than ours.

After the irradiation, the sample was kept at 10 K for one hour and was warmed up then with  $0.5\text{ K min}^{-1}$  to 293 K. Once the silane matrix has sublimed at about

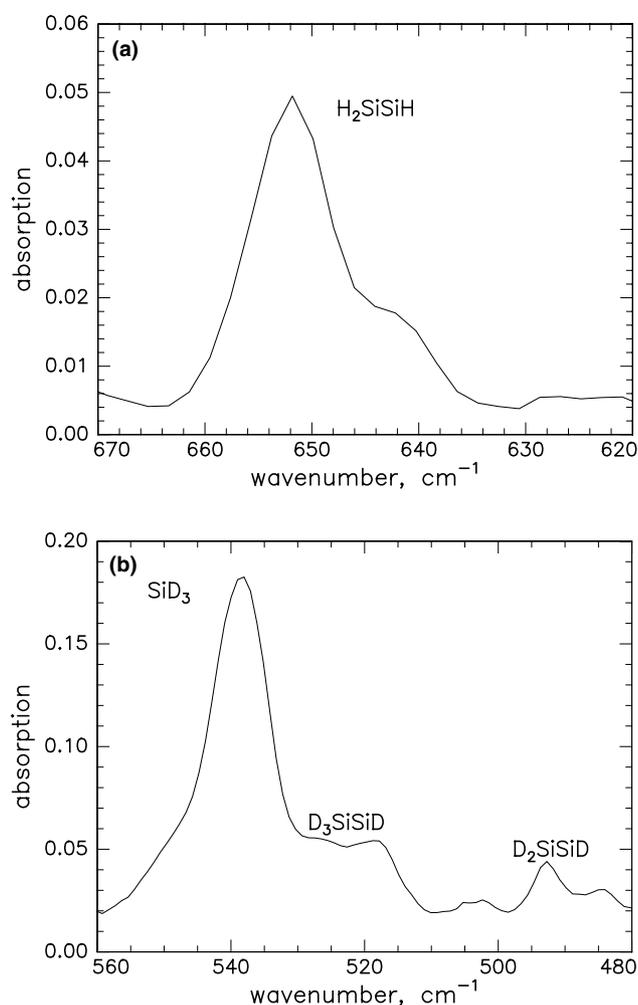


Fig. 3. New absorption features of the silylsilylenyl radical ( $\text{HSiSiH}_2$ ,  $X^2A$ ) isomer at  $651\text{ cm}^{-1}$  (a) and of the d3-silylsilylenyl radical ( $\text{DSiSiD}_2$ ,  $X^2A$ ) isomer (b) at  $493\text{ cm}^{-1}$  in the silane and d4-silane matrix at 10 K. The absorptions at  $528$  and  $540\text{ cm}^{-1}$  can be attributed to the  $\text{D}_3\text{SiSiD}$  and  $\text{SiD}_3$  species.

80 K, we detected additional peaks at  $898\text{ cm}^{-1}$  (silane experiment) and  $658\text{ cm}^{-1}$  (d4-silane experiment). These frequencies can be compared now with the calculated ones (Tables 2 and 3); we identify the  $898$  and  $658\text{ cm}^{-1}$  absorptions as the most intense  $\nu_{11}$  (symmetric  $\text{SiH}_2/\text{SiD}_2$  bending) fundamental of the thermodynamically most stable  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  isomer and its perdeuterated counterpart (Fig. 4).

## 5. Discussion and summary

Our combined experimental and theoretical investigation verify that the reaction sequence (1)–(3) to form the disilyl radical,  $\text{Si}_2\text{H}_5(X^2A)$ , is followed by a consecutive radiolysis of the disilyl radical to synthesize the silylsilylene isomer,  $\text{H}_3\text{SiSiH}(X^1A)$ , at 10 K (Fig. 5). This molecule and its perdeuterated isotopomers have

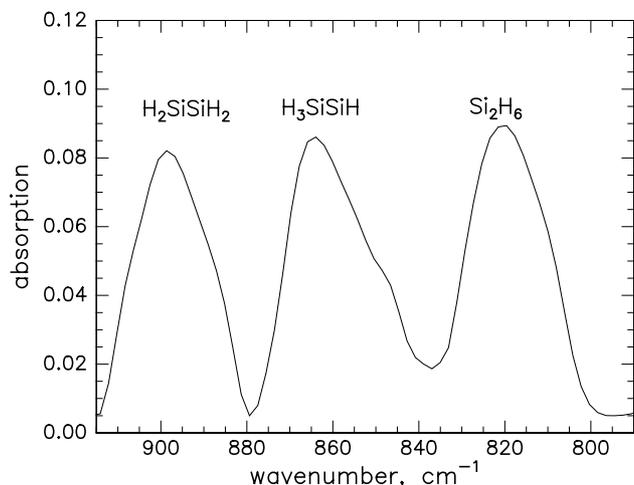


Fig. 4. Absorption features of the silylsilylene ( $\text{HSiSiH}_3$ ,  $X^1A'$ ) isomer at  $867\text{ cm}^{-1}$  and of disilene ( $\text{H}_2\text{SiSiH}_2$ ,  $X^1A_g$ ) at  $898\text{ cm}^{-1}$  after the sublimation of the silane matrix. The absorption of disilane at  $820\text{ cm}^{-1}$  is also shown.

been identified though their  $\nu_5$  fundamentals at  $867$  and  $635\text{ cm}^{-1}$ , respectively. In the warm up phase of the system after the actual irradiation, we are also able to observe the  $\nu_{11}$  absorptions of the thermodynamically most stable  $\text{Si}_2\text{H}_4$  isomer, disilene  $\text{H}_2\text{SiSiH}_2(X^1A_g)$ , and its isotopomer at  $898\text{ cm}^{-1}$  ( $\text{SiD}_4$  experiment) and  $658\text{ cm}^{-1}$  ( $\text{SiD}_4$  experiment). We try to answer now the question if both isomers are produced via an atomic hydrogen loss from the disilyl radical at  $10\text{ K}$  and the fundamentals of  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  are only obscured by the absorption of the silane matrix. Or is the silylsilylene species the only isomer being formed at  $10\text{ K}$ ? Integrating the  $\nu_5$  and  $\nu_{11}$  fundamentals and accounting for the integral absorption coefficients of these bands, we find a ratio of the  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  to  $\text{H}_3\text{SiSiH}(X^1A')$  of about one after the silane matrix has sublimed. Additionally, the number of  $\text{H}_3\text{SiSiH}(X^1A')$  species per  $\text{cm}^2$

sample does not decrease while the matrix is being annealed. These data suggest that both isomers are formed at  $10\text{ K}$ ; an isomerization of the  $\text{H}_3\text{SiSiH}(X^1A')$  isomer to the  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  structure can be likely ruled out since the intensity of the  $\nu_5$  fundamental does not change once the sample is warmed up. This interpretation gains also support from previous matrix studies of the  $\text{Si}_2\text{H}_4$  system [18]. Here,  $\text{H}_3\text{SiSiH}(X^1A')$  was found to isomerize to  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  only via a photon induced [1,2]-H-migration, but not thermally. The authors calculated also the thermal isomerization pathway. Actually, this process involves the bridged isomer,  $\text{H}_2\text{SiHSiH}(X^1A)$ , which was clearly not observed in our experiments. Also, a barrier of  $33\text{ kJ mol}^{-1}$  had to be overcome. This is certainly not feasible in our system: the infrared spectrum of both the  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  and  $\text{H}_3\text{SiSiH}(X^1A')$  isomers (Fig. 4) was taken at about  $70\text{ K}$ . This temperature corresponds to  $0.6\text{ kJ mol}^{-1}$ . Assuming a Maxwell–Boltzman distribution of the thermalized  $\text{H}_3\text{SiSiH}(X^1A')$  isomer at  $T = 70\text{ K}$ , less than 1% of the  $\text{H}_3\text{SiSiH}(X^1A')$  isomers can isomerize thermally to  $\text{H}_2\text{SiSiH}_2(X^1A_g)$ . Also, considering the differences of the enthalpies of formation of both  $\text{Si}_2\text{H}_4$  isomers of  $27 \pm 5\text{ kJ mol}^{-1}$ , we would expect a  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  to  $\text{H}_3\text{SiSiH}(X^1A')$  ratio of about  $10^{19}$ . This is clearly not the case. Therefore, we must conclude that in our studies, both experimentally observed  $\text{Si}_2\text{H}_4$  isomers are formed via non-equilibrium, suprathreshold processes at  $10\text{ K}$  through a silicon–hydrogen bond rupture within the disilyl radical. We would like to stress that we averaged the infrared spectra over  $180\text{ s}$  each. Therefore, we cannot exclude that to a minor amount  $\text{H}_2\text{SiSiH}_2(X^1A_g)$  and  $\text{H}_3\text{SiSiH}(X^1A')$  are formed via concerted four center [1,2] and three center [1,1] molecular hydrogen elimination pathways. Future experiments employing  $\text{H}_3\text{SiSiD}_3$  will solve this remaining question.

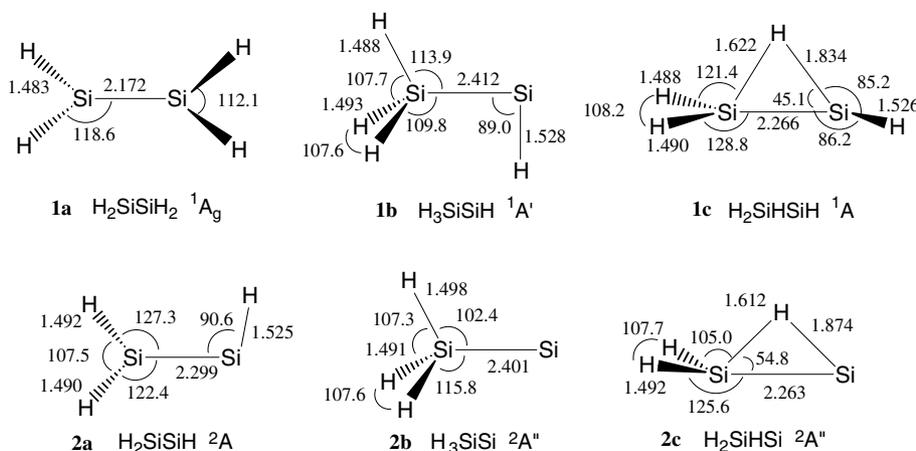


Fig. 5. Optimized structures of  $\text{Si}_2\text{H}_4$  and  $\text{Si}_2\text{H}_3$  species calculated with the B3LYP/6-311G(d,p) method. The bond lengths and angles are given in units of Å and  $^\circ$ , respectively.

Upon further irradiation of the sample, the  $\text{Si}_2\text{H}_4$  isomers were found to be radiolyzed to the  $\text{Si}_2\text{H}_3$  radical  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$ ; the latter has been assigned via the  $\nu_5$  fundamental. Neither the  $\text{H}_2\text{SiHSi}(\text{X}^2\text{A}'')$  nor the  $\text{H}_3\text{SiSi}(\text{X}^2\text{A}'')$  isomers could be identified since their absorptions overlap with the silane matrix and with the disilyl radical. Even upon a sublimation of the matrix, the elusive  $\text{Si}_2\text{H}_3$  structures could not be detected. Recall that the  $\text{H}_2\text{SiSiH}_2(\text{X}^1\text{A}_g)$  and  $\text{H}_3\text{SiSiH}(\text{X}^1\text{A}')$  precursors were formed in a 1:1 ratio. A simply Si–H bond rupture in  $\text{H}_2\text{SiSiH}_2(\text{X}^1\text{A}_g)$  would lead solely to the experimentally observed  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  isomer. The  $\text{H}_3\text{SiSiH}(\text{X}^1\text{A}')$  structure has two non-equivalent silicon–hydrogen bonds. If the Si–H bond is cleaved statistically, we expect a  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  to  $\text{H}_3\text{SiSi}(\text{X}^2\text{A}'')$  ratio of 3:1. These thoughts postulate a dominant formation of  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  as verified experimentally. The present study completes the gap in the infrared spectroscopic characterization of small hydrogenated silicon clusters  $\text{Si}_2\text{H}_y$  ( $y = 3\text{--}6$ ). The new infrared absorption of the  $\text{H}_2\text{SiSiH}(\text{X}^2\text{A})$  radical may be employed in future spectroscopic investigations of chemical vapor deposition processes and in astronomical searches in the infrared regime toward the carbon star IRC + 10216.

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### References

- [1] R. Prasad, N. Chakraborti, *Met. Mater. Proc.* 10 (1998) 203.
- [2] P. Ho, M.E. Coltrin, J.S. Binkley, C.F. Melius, *J. Phys. Chem.* 90 (1986) 3399.
- [3] M.C. Ernst, A.F. Sax, J. Kalcher, *Chem. Phys. Lett.* 216 (1993) 189.
- [4] M.T. Swihart, *J. Phys. Chem. A* 104 (2000) 6083.
- [5] X.G. Gong, D. Guenzburger, E.B. Saitovitch, *Chem. Phys. Lett.* 275 (1997) 392.
- [6] C. Pak, J.C. Rienstra-Kiracofe, H.F. Schaefer, *J. Phys. Chem. A* 104 (2000) 11232.
- [7] G.R. Gupte, R. Prasad, *Int. J. Mod. Phys. B* 12 (1998) 1607.
- [8] T. Shirafuji et al., *Jpn. J. Appl. Phys.* 32 (1993) 1546.
- [9] J.G.E. Gardeniers, L.J. Giling, F. De Jong, J.P. van der Eerden, *J. Cryst. Growth* 104 (1990) 727.
- [10] S.L. Girshick, M.T. Swihart, S.M. Shu, M.R. Mahajan, S. Nijhawan, *Proc. Electrochem. Soc.* 98 (1999) 215.
- [11] M.T. Swihart, S.L. Girshick, *J. Phys. Chem. B* 103 (1999) 64.
- [12] A. Matsuda, K. Tanaka, *Thin Solid Film* 92 (1982) 172.
- [13] Z.X. Wang, M.B. Haung, *J. Chem. Soc., Faraday Trans.* 94 (1998) 635.
- [14] S. Oikawa, M. Tsuda, S. Ohtsuka, *J. Mol. Struct.* 310 (1994) 287.
- [15] B. Ruscic, J. Berkowitz, *J. Chem. Phys.* 95 (1991) 2416.
- [16] L. Andrews, X. Wang, *J. Phys. Chem. A* 106 (2002) 7696.
- [17] D. Sillars, C. Bennett, R.I. Kaiser, Y. Osamura, *Chem. Phys.* 2004 (in press).
- [18] G. Maier, H.P. Reisenauer, J. Glatthaar, *Chem. Eur. J.* 8 (2002) 4383.
- [19] L. Sari, M.C. McCarthy, H.F. Schaefer, P. Thaddeus, *J. Am. Chem. Soc.* 125 (2003) 11409.
- [20] B.E. Turner, *Astrophys. J.* 376 (1991) 573.
- [21] R.I. Kaiser, *Chem. Rev.* 102 (2002) 1309.
- [22] D.M. Goldhaber, A.L. Betz, *Astrophys. J.* 279 (1984) L55.
- [23] C. Bennett, A.M. Mebel, R.I. Kaiser, *Phys. Chem. Chem. Phys.*, 2004 (submitted).
- [24] W.M. Sears, J.A. Morrison, *J. Chem. Phys.* 62 (1975) 2736.
- [25] A.D. Becke, *J. Chem. Phys.* 98 (1993) 5648.
- [26] C. Lee, W. Yang, R.G. Parr, *Phys. Rev. B* 37 (1988) 785.
- [27] R. Krishnan, J.S. Binkley, R. Seeger, J.A. Pople, *J. Chem. Phys.* 72 (1980) 650.
- [28] N. Chakraborti, P.S. de, R. Prasad, *Z. Metallkunde* 90 (1999) 7.
- [29] J. Cizek, *Adv. Chem. Phys.* 14 (1969) 35.
- [30] J.A. Pople, M. Head-Gordon, K. Raghavachari, *J. Chem. Phys.* 87 (1987) 5968.
- [31] R.A. Kendall, T.H. Dunning, R.J. Harrison, *J. Chem. Phys.* 96 (1992) 6796.
- [32] M.J. Frisch et al., *GAUSSIAN 98 Revision A11*, Gaussian, Pittsburgh, PA, 2002.
- [33] L. Andrews, X. Wang, *J. Phys. Chem. A* 106 (2002) 7696.
- [34] L. Li, J.T. Graham, W. Weltner, *J. Phys. Chem. A* 105 (2001) 11018.
- [35] N. Legay-Sommaire, F. Legay, *J. Phys. Chem. A* 102 (1998) 8759.